New Vacuum Processes Achieve Mechanical Property Improvement in Gearbox Components

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Management Summary

This paper introduces new process developments in low-pressure carburizing and carbonitriding using either high-pressure gas quenching or interrupted gas quenching. (In this article, the interrupted gas quenching method used is StopGQ®, a registered trademark of ECM.) Comparison of mechanical properties which result from each process will be discussed. The reader will discover how the optimization of carbon and nitrogen enrichment and the mastering of gas quenching can improve fatigue strength and impact properties.

Introduction

Increased performance requirements in automotive and other industries demand higher strength characteristics from components such as gears and pinions. Those requirements translate into a need for improved impact properties and fatigue strength. The technical solution to address the need for expanding the performance envelope has required the development and optimization of new processes.

A research program was launched by ECM using the flexibility of lowpressure vacuum carburizing in combination with high-pressure gas quenching (LPC + HPGQ) to investigate several new vacuum processes:

- LPC + HPGQ + StopGQ quenching
- LPCN (low-pressure carbonitriding) + HPGQ
- LPC + HPGQ + StopGQ quenching

Marked improvements in mechanical properties were found as a result of the optimization of both surface enrichment techniques and quench parameters. The conventional process, LPC + HPGQ + tempering, was used as a benchmark for all tests.

StopGQ Quenching Concept

Once a workload has been hardened or case hardened, tempering is necessary. Conventional high-pressure gas quenching looks to bring the entire workload to room temperature as rapidly as possible. Interrupted quenching involves halting the cooling process in the temperature range of 350-400°F (180-200°C) and introducing an isothermal hold (Fig. 1) in order to perform an "auto-tempering" step in the gas quenching cell, thus avoiding the need for subsequent tempering. Instrumented full-load trials of gears (Fig. 2) helped determine the correct time delay before initiation of quench interruption as a function of quench pressure.

LPCN Studies

Atmosphere carbonitriding is typically performed in the temperature range of $1,475-1,650^{\circ}F$ (800-900°C). Typical case depths are $0.010-0.020^{"}$ (0.25-0.50 mm), although deeper and shallower cases can be achieved.



Figure 1—StopGQ quench concept.



Figure 2—StopGQ quench time versus pressure relationship.



Figure 3—LPCN boost/diffuse steps.

Surface carbon is normally in the range of 0.6-0.8%, and surface nitrogen content is in the range of 0.15-0.30%.

The principle of low-pressure vacuum carbonitriding is to alternate the boost/diffuse gas mixtures between hydrocarbon gas (propane or acetylene) and nitrogen or hydrocarbon gas and ammonia (Fig. 3). Processing pres-



Figure 4—LPCN carbon and nitrogen profiles.

Table 1—Results								
	Carbonitriding Temperature, °F (°C)	Carbon (%)	Nitrogen (%)	Effective Case Depth, Inches (mm)	Microstructure			
Target Values		0.6–0.8	0.15– 0. 30	650 HV (58HRC) @ 0.020" (0.5 mm)				
	1700 (930)	0.88	0.13	0.024 (0.62)	Martensite + Austenite (<20%)			
	1750 (960)	0.80	0.14	0.040 (1.00)	Martensite + Retained Austenite (<20%)			



Figure 5—LPCN elevated temperature test results.

Table 2—Specification targets for test gears							
Steel Grade Effective Case Depth @ 52.5 HRC (550 HV) Inches (mm)		Surface Hardness (HV 20)	Core Hardness (HV 50)				
5130	0.020-0.030 (0.5-0.7)	690–900	320–500				



Figure 6—Gear test load.

& Nitrogen Profiles sure was equivalent to LPC—that is, in the range of 4–15 torr (5–20 mbar). Ammonia is added during the later boost/diffuse steps and during the final soak. The amount and duration of the carbonitriding steps depends on the depth and nitrogen concentration desired. Trials (Fig. 4) were conducted on SAE 5130 (29MnCr5) pinions at 1,615°F (880°C), followed by direct quenching using 8-bar nitrogen.

on SAE 5130 (29MnCr5) pinions at 1,615°F (880°C), followed by direct quenching using 8-bar nitrogen. Samples used were placed in the top, center and bottom baskets of the load. Analysis of carbon and nitrogen content involved two different technologies: GDOS (Glow Discharge Optical Emission Spectrometry) to measure the nitrogen concentration of the surface to a depth of 0.0015" (40 µm) and WDS (Wavelength Dispersive X-Ray Spectroscopy), accurate to a depth of up to 0.060" (1.5 mm). These techniques confirmed that the profiles using LPCN were equivalent to what was produced by atmosphere carbonitriding.

Additional LPCN tests (Fig. 5) were conducted at higher temperatures to investigate cycle time savings. Typical carburizing temperatures of 1,700°F (930°C) and 1,760°F (960°C) were selected for study. Process parameters were similar to the trials at 1,615°F (880°C). GDOS profiles were conducted to determine the nitrogen content at the near surface up to 0.0012'' (30 µm). Results indicated a nitrogen content of 0.5%, falling rapidly to 0.15% below the near surface at 1,700°F (930°C) and the same behavior was observed at 1760°F (960°C). The nitrogen profile was achieved at 1,700°F (930°C) up to 0.015" (0.4 mm) and up to 0.40" (1.0 mm) at 1,760°F (960°C). These tests indicate that the process can work at high temperatures as well.

Fatigue and Impact Studies

Specification targets (Table 2) were selected for gears (Fig. 6) of SAE 5130 (29MnCr5) material.

Hardness and effective case depth results (Table 3) achieved targeted values. This confirmed that there was no metallurgical difference to influence fatigue strength results.

One of the mechanical property tests employed was to determine impact properties. Impact samples (Fig. 7) were tested in a pendulum type impact tester at 50J.

The result of impact testing when compared to LPC + HPGQ revealed, as one might expect, a strong benefit of tempering on impact properties. Noteworthy is the improvement over LPC + HPGQ achieved by interrupted quenching or by LPCN + HPGQ. These results indicate that additional testing is required to optimize results. The use of LPCN + HPGQ with StopGQ quenching (Fig. 8) resulted in impact values exceeding those of LPC + HPGQ + tempering.

Another mechanical test employed was that of rotating bending fatigue involving a notched sample geometry (Fig. 9). The test was run for 1×10^7 cycles.

The results of rotating bending fatigue testing (Fig. 10) indicate all of the new processes improve strength values over those of either LPC + HPGQ or LPC + HPGQ + tempering.

Realized Objectives

An improvement in fatigue strength was realized using the "auto-tempering" effect achieved by StopGQ quenching. The effect of nitrogen present in the surface layer of LPCN parts was revealed in higher impact and fatigue strength values (Fig. 11).

Table 3—Hardness and Case Depth Summary								
Steel	Thermal	Hardr	ness	Case Depth,				
Grade	Treatment	(H)	/)	Inches (mm)				
		Surface	Core					
5130	LPC + HPGQ	885	440	0.015 (0.39)				
5130	LPC + HPGQ +Tempering	800	445	0.015 (0.40)				
5130	LPC + HPGQ with StopGQ Quenching	872	460	0.018 (0.45)				
5130	LPCN + HPGQ	927	460	0.015 (0.38)				
Note: Quench pressure was 13 bar (nitrogen) for each test run.								



Figure 7—Impact sample.



Figure 8-Improvement of impact properties over LPC+HPGQ by process.



Figure 9—Rotating bending fatigue sample.



Figure 10-Improvement of rotating beam fatigue properties over LPC+HPGQ by process.



Figure 11a—Summary of StopGQ® benefits versus LPC+HPGQ.







Figure 11c—Summary of LPCN benefits versus LPC+HPGQ +

Dilatometry studies (Fig. 12) showed less contraction with an interrupted quench compared to direct high-pressure gas quenching. During the StopGQ quench, quadratic martensite is transformed into cubic martensite plus ε carbides, resulting in an automatic tempering effect. This was demonstrated by rupture analysis (Fig. 13), which revealed higher ductility in the core of the material processed with LPCN + StopGQ quenching.

Scanning electron microscopy (SEM) analysis at 3,700X comparing LPC and LPCN microstructures found fine precipitates of carbonitrides in the latter, which are believed to have a strong influence on fatigue strength (Fig. 14).

These trials allowed the following conclusions to be reached:

- Conditions were established to predict and control carbon and nitrogen concentration profiles.
- The metallurgical parameters which act on resistance in fatigue inflection of gear teeth for a fixed hardened depth were better understood.
- Results indicate improvements in fatigue resistance of gear teeth compared to low-pressure vacuum carburizing treatments applied to gear boxes.

Future Studies

Future research and development efforts will target further understanding of the role of nitrogen on mechanical properties. Those efforts will include microstructural analysis of grain boundaries to determine if the grain size has been refined. In addition, further optimization of the influence of StopGQ quenching temperature and hold time will be investigated.

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Figure 12-Dilatometry studies of the StopGQ quenching method.



Figure 13-Rupture analysis comparison by process.



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